

High Performance Photovoltaic Applications Using Solution-Processed Small Molecules

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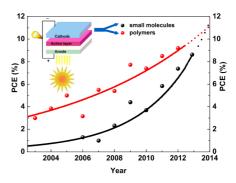
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RECEIVED ON MARCH 28, 2013

CONSPECTUS

E nergy remains a critical issue for the survival and prosperity of human civilization. Many experts believe that the eventual solution for sustainable energy is the use of direct solar energy as the main energy source. Among the options for renewable energy, photovoltaic technologies that harness solar energy offer a way to harness an unlimited resource and minimum environment impact in contrast with other alternatives such as water, nuclear, and wind energy.

Currently, almost all commercial photovoltaic technologies use Si-based technology, which has a number of disadvantages including high cost, lack of flexibility, and the serious environmental impact of the Si industry. Other technologies, such as organic photovoltaic (OPV) cells, can overcome some of



these issues. Today, polymer-based OPV (P-OPV) devices have achieved power conversion efficiencies (PCEs) that exceed 9%. Compared with P-OPV, small molecules based OPV (SM-OPV) offers further advantages, including a defined structure for more reproducible performance, higher mobility and open circuit voltage, and easier synthetic control that leads to more diversified structures. Therefore, while largely undeveloped, SM-OPV is an important emerging technology with performance comparable to P-OPV.

In this Account, we summarize our recent results on solution-processed SM-OPV. We believe that solution processing is essential for taking full advantage of OPV technologies. Our work started with the synthesis of oligothiophene derivatives with an acceptor—donor—acceptor (A-D-A) structure. Both the backbone conjugation length and electron withdrawing terminal groups play an important role in the light absorption, energy levels and performance of the devices. Among those molecules, devices using a 7-thiophene-unit backbone and a 3-ethylrhodanine (RD) terminal unit produced a 6.1% PCE. With the optimized conjugation length and terminal unit, we borrowed from the results with P-OPV devices to optimize the backbone. Thus we selected BDT (benzo[1,2-b:4,5-b']dithiophene) and DTS (dithienosilole) to replace the central thiophene unit, leading to a PCE of 8.12%. In addition to our molecules, Bazan and co-workers have developed another excellent system using DTS as the core unit that has also achieved a PCE greater than 8%.

34 Introduction

The solar cell has been considered a green and effective 35 technology to address today's energy and environment 36 issues. Though conventional solar cells based on silicon 37 and other inorganic materials have come into commercial-38 ization, there are still some rather challenging issues to be 39 solved to address their high cost, limited flexibility and 40 environmental issues.¹ With the advantage of low cost, 41 lightweight, flexibility, and so forth, organic photovoltaics 42 (OPV) has been regarded as one of the promising alterna-43 tives to overcome those issues.² 44

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Currently, OPV devices are based on two types of elec-45 tron donor materials, polymers and small molecules. In the 46 past 20 years, polymer-based solution processed bulk het-47 erojunction (BHJ)³ solar cells have been dominant and a 48 power conversion efficiency (PCE) over 9% has been 49 achieved⁴ for single layer devices since it seemed that small 50 molecule donors could not form a good phase-separated 51 mixture with the acceptors such as [6,6]-phenyl-C₆₁-butyric 52 acid methyl ester (PC₆₁BM) or [6,6]-phenyl-C₇₁-butyric acid 53 methyl ester (PC71BM) owing to their rigid structures and 54 commonly bad film-forming properties by solution processing, 55

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and thus low PCEs are obtained. Therefore, most of the small
 molecule-based OPV (SM-OPV) devices have been focusing on
 the vacuum evaporation approach to form planar heterojunc tion or bulk heterojunction devices,⁵ although they have a
 higher cost than the solution processing method.

61 Though great progress has been made recently for the polymer based OPV (P-OPV) devices, recent studies indicate 62 that the intrinsic batch to batch variation issue of polymers 63 due to different polymerization degrees and polydispersity 64 index⁶ could hamper the performance reproducibility for 65 possible commercialization of OPV. In contrast, SM-OPV 66 devices demonstrate many prominent advantages, that is, 67 (1) less batch to batch variation for performance due to their 68 uniform and defined molecular structures; (2) easily tuned 69 absorptions and energy levels resulting from a dedicated 70 chemical structure design; (3) a generally intrinsic higher 71 mobility and open circuit voltage (V_{oc}). These fundamental 72 reasons and recent significant developments indicate that 73 SM-OPV devices could play a much bigger role for OPV 74 75 devices in general and are likely to achieve the same or even better performance than P-OPV devices.^{7,8} 76

77 From the equation for

$$PCE = (J_{sc} \times V_{oc} \times FF)/P_{in}$$
(1)

PCEs are decided by three factors (Figure 1) under a given F1 78 incident light power density (Pin): short circuit current 79 density (J_{sc}) , open circuit voltage (V_{oc}) , and the fill factor 80 (FF). It is becoming clearer that the mechanism behind this 81 equation is much more complicated and the overall 82 performance of OPV devices depends on many and in 83 some cases competing factors,⁶ which can be cataloged 84 into four levels. The first is the intrinsic molecular struc-85 tures and energy level of the donor and acceptor materi-86 als, which have fundamental impact on the absorption 87 thus J_{sc} energy level thus V_{oc} and so forth.⁹ The ideal 88 case is to have the strongest absorption over the whole 89 solar spectrum and a maximum in the energy difference 90 between the LUMO of the acceptor and the HOMO of the 91 donor. The second is the morphology/structure control of 92 the active layer. This has great influence on FF and 93 in many cases J_{sc} and V_{oc} too, primarily due to the 94 differences in exciton diffusion, charge separation/ 95 recombination, and transport step efficiencies. Note that 96 recent developments using solvent/thermal annealing 97 and additives has greatly increased PCEs for the same 98 active materials using this approach.^{10,11} The third is the 99

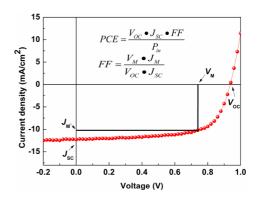


FIGURE 1. Typical J-V curve of an OPV device indicating the meaning of J_{sc} , V_{oc} , FF, and PCE. The current density and voltage at the maximum power point are denoted as J_{M} and V_{M} , respectively.

device structure/optimization including using interface 100 layer, which can further optimize the V_{oc} and J_{sc} by adjusting 101 the work function of charge collecting electrodes, the inter-102 nal interface electrical field etc.⁴ Indeed, this optimization has 103 improved the device performance significantly for the same 104 materials.^{4,12} The fourth is that, for practical applications, the 105 stability (degradation of the PCE) and the lifetime of the 106 devices, which has attracted relatively less attention.^{12,13} 107

Like polymer-based OPV devices, in addition to other all 108 important issues such as device optimization and morphol-109 ogy control as discussed above, the first fundamental issue is 110 to design optimized small molecules as the donor materials 111 with the required solar absorption and energy levels. Learn-112 ing from recent developments of both polymer and small 113 molecule-based OPV devices, it is believed that several 114 requirements need to be satisfied simultaneously for these 115 small molecules to have the advantages of both conven-116 tional small molecules (rigid small molecules with short or 117 no conjugation, such as phthalocyaine, triphenyl amine, etc.) 118 and polymers. These include (1) excellent film formation 119 ability, which can be achieved by controlling the appropriate 120 conjugation length and alkyl groups; (2) wide and efficient 121 absorption, which can be achieved using the appropriate 122 conjugation length (~6-10 conjugated units) and good 123 photosensitive groups; (3) matched energy levels with the 124 acceptors, achievable by electron rich/poor conjugation unit 125 and substituted groups; (4) high charge mobility, generally 126 an advantage of small molecules, especially of those with a 127 planar molecular structure; and (5) good solubility, which 128 can be achieved by using long and/or branched alkyl chains. 129

Bearing these in mind and based on some of the pioneering studies for small molecules using the vacuum deposition approach,^{14,15} we have developed a series of small molecules with an acceptor–donor–acceptor (A-D-A) structure,^{8,16–27} 133

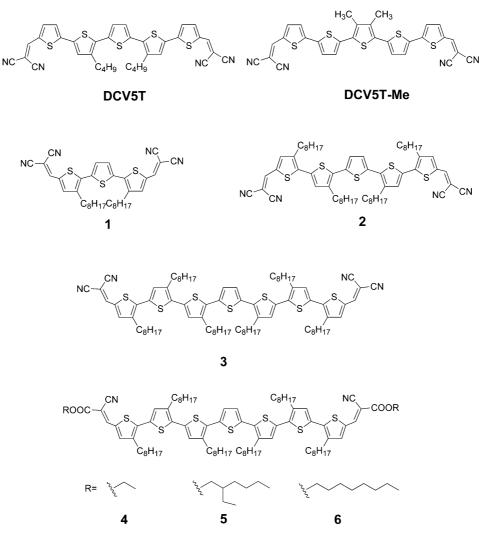


FIGURE 2. Chemical structures of DCV5T, DCV5T-Me, and molecules 1-6.

which consists of conjugated donor backbone main chain and 134 acceptor terminal units. PCEs over 8% have been achieved by 135 using the simple solution process,⁸ which is comparable with 136 that of the best P-OPV devices.⁴ In the past few years, there 137 have been some excellent and comprehensive reviews for all of 138 the four levels mentioned above.^{5,28,29} In this Account, we will 139 focus on the chemical viewpoint of small molecule donor 140 material design and its impact on the overall performance for 141 SM-OPV devices using fullerene derivatives as the acceptors, 142 with some brief discussion for the morphology control and 143 device optimization. Some structure-performance relationships 144 and general rules will also be proposed, which may offer some 145 clues for future material and device optimization. 146

Septithiophene Based Molecules with Electron Withdrawing Terminal Units

149 It is not surprising that oligothiophenes have been widely150 studied for OPV applications since the thiophene unit has

always been thought to be one of the best building blocks for 151 most organic functional materials.³⁰ In 2006, Roncali et al. 152 reported a series of oligothiophene derivatives, where a 153 tetrahedral silicon core-based three-dimensional oligothio-154 phene was used as the donor for a solution-processed SM-155 OPV device with PC₆₁BM, and a PCE of 0.3% was obtained.³¹ 156 Later on, other branched or star-shaped oligothiophenes 157 were designed by Roncali and other groups and PCEs below 158 2% were obtained.^{32,33} The performance is hampered by 159 the limited absorption in the visible and near-infrared region 160 for these oligothiophenes due to the relatively large band 161 gap (>2.2 eV). It is therefore necessary to increase the 162 conjugation length to the generally accepted effective con-163 jugation length ($\sim 6-10$ units) in polymers such as in P3HT. 164 To adjust the band gap, another strategy is to use the 165 donor-acceptor approach. Importantly, in 2006, Bäuerle 166 et al. reported a small molecule DCV5T (Figure 2) by intro-167 F2 ducing the strong electron withdrawing unit dicyanovinyl 168

(DCV) into quinquethiophene derivatives for a vacuum
evaporation OPV device.¹⁴ A PCE of 6.9% has been achieved
very recently based on a DCV5T analogue molecule DCV5TMe (Figure 2).¹⁵

Based on these previous studies and with the goal to have 173 the required conjugation length and also be able to use 174 solution processing, our work started with the design and 175 synthesis of one-dimensional oligothiophene derivatives 176 (1-3) substituted with long alkyl groups and the DCV 177 terminal unit.¹⁶ The UV-vis absorption spectra showed that 178 the introduction of DCV electron-accepting units resulted in 179 not only a large red shift of the absorption but also higher 180 absorption coefficients compared with the nonsubstituted 181 oligothiophenes. The optical band gaps from the solution 182 absorption (Figure 3) of these A-D-A molecules become **F3**183 significantly smaller with increasing the number of thio-184 phene units for compounds 1-3, with values of 1.90, 1.74, 185 and 1.68 eV, respectively. Thus, molecule 3 was used as the 186 donor to fabricate solution processed BHJ cells with PC₆₁BM 187 and a PCE of 2.45% was obtained.¹⁷ Further device optimi-188 zation gave a PCE of 3.7% with a high J_{sc} of 12.4 mA/cm², V_{oc} 189 of 0.88 V, but a low FF of 0.34.18 190

The low FF for compound **3** is believed to be due to the 191 poor film quality caused by the high rigid planarity and low 192 solubility, though it is much improved compared with other 193 194 conventional small molecules. Thus, we introduced an electron-withdrawing alkyl cyanoacetate group to replace DCV 195 as the terminal unit in the septithiophene backbone and 196 synthesized three molecules 4-6 (Figure 2).¹⁹ Indeed, the 197 OPV devices based on these molecules and PC₆₁BM demon-198 strated significantly higher PCEs than that of 3, with a much 199 improved FF (\sim 0.50) and similar V_{oc} and J_{sc} as expected. 200 Note the optimized weight ratio of D:A is around 1:0.5, 201 which is very different from that (1:3-4) of many optimized 202 polymer based OPV devices.³⁴ This result, which was also 203 observed by other groups³⁵ and in most of our SM-OPV 204 devices,^{8,16–21,23–27} could be an important advantage. The 205 much improved FF is supported by the better interpenetrat-206 ing network morphology which would promote more effi-207 cient charge separation and transport.⁵ 208

Note that the best J_{sc} is over 20 mA/cm² for polymers³⁶ 209 and the value is relatively smaller for small molecules, so we 210 introduced some dye building blocks to obtain stronger solar 211 212 absorption. Thus, a series of dye building blocks including 3-ethylrhodanine (RD) were introduced into the septithio-213 phene backbone (Figure 4).^{20–22} Overall, all these com-**F4**214 pounds, 7-12, exhibit an obvious improvement in the 215 absorption with red-shift and higher absorption coefficients. 216

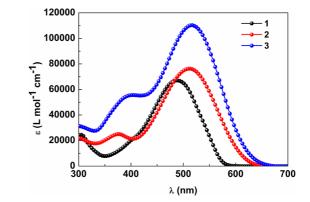


FIGURE 3. UV-vis spectra of compounds $1-3 (10^{-5} \text{ mol } L^{-1})$ in chloroform.

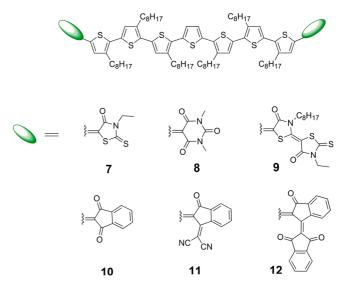


FIGURE 4. Chemical structures of molecules 7-12.

Systematical investigation indicates that compound 7 gave the 217 best result with a record PCE of 6.10% with a high $V_{\rm oc}$ of 0.92 V 218 and a remarkable J_{sc} of 13.98 mA/cm². The improved J_{sc} was 219 supported by the external quantum efficiency (EQE) curve 220 (Figure 5) with a broad response covering 350–750 nm, with 221 F5 maximum nearly 70%. The comparison of the EQE data 222 (Figure 5) of compound 7 and previous best alkyl cyanoacetate 223 group terminated compound 6 also supports our strategy of in-224 creasing the photo-response by introducing dye building blocks. 225

Note that most of the devices using these molecules 226 exhibit PCEs in the range of 4-6%, which indicate the 227 backbone is a good choice. But it has been proved that 228 elegant design is needed to master and balance all the 229 factors to achieve the best performance.^{21,22} The absorp-230 tion, energy level and device performance of these mole-231 cules together with those of 6 and 7 are summarized in 232 Table 1. The absorptions of these molecules in solution and 233 T1 on solid films differ with different terminal units owing to 234

their different electron withdrawing abilities and molecule 235 packing modes. The HOMOs and LUMOs are also affected 236 by the terminal units, especially the LUMO level (Table 1), 237 with compound 11 being a good example. In addition, 238 though with six octyl chains on the septithiophene back-239 bone, the terminal unit still plays a great role for their 240 solubility. For example, molecule 6 with an n-octyl cyanoa-241 cetate terminal unit has the largest solubility in CHCl₃ 242 (204 mg/mL), while 2-(3-oxo-2,3-dihydroinden-1-ylidene) 243 malononitrile (compound 11) reduces the solubility substan-244 tially to 4.6 mg/mL, too low to fabricate a device using the 245 solution process. While these molecules exhibit generally 246 similar $V_{\rm oc}$ due to the same backbone structure, their $J_{\rm sc}$ and 247 FF change significantly due to different end groups, probably 248 mainly due to the different molecule packing and morphol-249 ogy in the active layer. Overall, alkyl cyanoacetate and RD, 250 especially the latter with the highest $V_{\rm oc}$ and $J_{\rm sc}$ are the best 251 terminal units among the end units we tested for this series 252 of molecules.^{19–22} Also, compound **10** with a 1,3-indane-253 dione end unit gave the best FF of 0.72, which is not only the 254 highest in the series of small molecule donors but also 255 among the highest FFs for P-OPV devices.^{4,34} 256

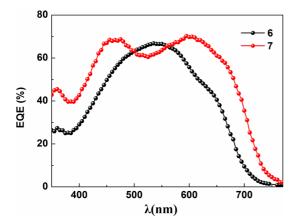


FIGURE 5. EQE of BHJ devices derived from a blend of 6/PC₆₁BM and **7**/PC₆₁BM with D:A weight ratio of 1:0.5.

Obviously, there is a question whether the optimized 257 length of the backbone structure is seven units or not in 258 the above oligothiophene systems. Quinquethiophenes 259 have been proved to have a deeper HOMO level than 260 septithiophenes for the decreased conjugations, and there-261 fore, a higher V_{oc} might be expected.¹⁴ Thus, three repre-262 sentative compounds of 13-15 with five units of thiophene 263 and three optimized end groups were synthesized and 264 investigated for their OPV performance (Figure 6, Table 1).²³ 265 F6 Among them, 14 indeed shows a higher $V_{\rm oc}$ of 1.08 V as 266 expected but with only a moderate PCE of 4.63%, but 13 and 267 15 showed an unexpected lower $V_{\rm oc}$ (0.88 and 0.78 V) than 268 the corresponding septithiophene derivatives and lower 269 PCEs of 3.27% and 4.00%, respectively. The lower $V_{\rm oc}$ for 270 13 and 15 could be due to the stronger donor-acceptor 271 interaction.³⁷ The hole mobility of these three compounds is 272 similar in magnitude to their septithiophene analogues. 273 However, their J_{sc} values are all significantly lower than 274 those of the corresponding septithiophene analogues, par-275 tially corresponding to their lower light absorption effi-276 ciency. The FF is generally similar or larger for these 277 molecules (13–15) compared to compound 7, probably 278 due to their better planarity and therefore more efficient 279 packing. Compound 16 with a longer conjugation backbone 280 but using a benzothiadiazole unit as the core and quin-281 quethiophene as the linkage only gave a moderate of PCE 282 (3.07%).²⁴ From the performance of above molecules, it 283 seems that molecules with a backbone of septithiophene 284 (7 thiophene units) exhibit better performance in the above 285 oligothiophene-based molecules. 286

Modifying the Main Oligothiophene Backbone 287

With the above systematic studies for oligothiophenes and 288 particularly the septithiophene derivatives, we turned our 289 attention to the modification of the backbone unit. Thus, two 290 of the best units, BDT (benzo[1,2-b:4,5-b']dithiophene)³⁸ and 291 DTS (dithienosilole)³⁹ widely used in P-OPV devices, have 292

TABLE 1. Absorption, Energy Levels, and Photovoltaic Properties of Molecules 6–16 under the Illumination of AM 1.5G, 100 mW/cm ²													
SM	$\lambda_{\max}^{\text{film}}$ (nm)	$E_{\rm g}^{ m opt}$ (eV)	$E_{\rm g}^{\rm cv}$ (eV)	HOMO (eV)	LUMO (eV)	solubility (mg/mL)	$\mu_{\rm h} imes 10^{-4} ({\rm cm}^2 {\rm V}^{-1} {\rm s}^{-1})$	$V_{\rm oc}$ (V)	J _{sc} (mA/cm ²)	FF	PCE (%)	ref	
6 7 8 9 10	580 618 605 584 630, 692	1.74 1.69 1.67 1.70 1.49	1.84 1.72 1.62 1.70 1.53	-5.13 -5.00 -5.12 -5.09 -4.97	-3.29 -3.28 -3.50 -3.39 -3.44	204 104 29 32 40	3.26 1.50 0.47 0.24 1.73	0.86 0.92 0.90 0.92 0.80	10.74 13.98 7.54 6.77 8.56	0.55 0.474 0.60 0.39 0.72	5.08 6.10 4.05 2.46 4.93	19 20 21 21 22	
11 12 13 14 15 16	718, 802 684 561 615, 671 622, 680 645	1.33 1.20 1.80 1.65 1.56 1.60	1.30 1.04 2.02 1.89 1.75 1.77	5.02 4.90 5.25 5.09 5.11 4.98	-3.72 -3.86 -3.23 -3.20 -3.36 -3.21	4.6 17	0.30 3.94 5.76 5.51 1.3	0.76 0.88 1.02 0.78 0.78	3.14 7.02 9.26 8.13 7.10	0.28 0.53 0.49 0.63 0.554	0.66 3.27 4.63 4.00 3.07	22 22 23 23 23 23 24	

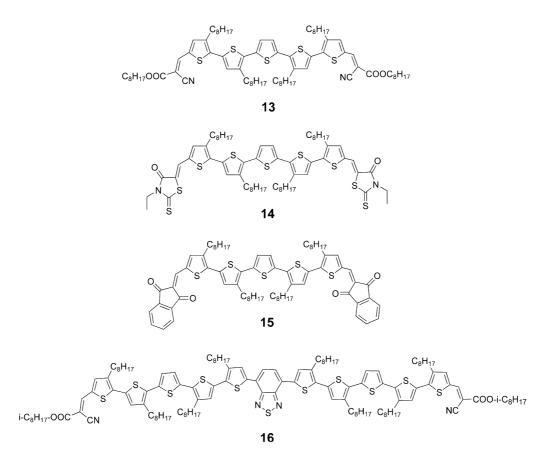


FIGURE 6. Chemical structures of molecules 13–16.

F7293 been used to replace the central thiophene unit in the **T2**294 septithiophene system (Figures 7 and 9, Table 2).

295 BDT Core Based Small Molecules

This study started with an unsubstituted BDT system 296 (compound 17).²⁵ It turned out that the absorption and 297 energy level of 17 were comparable with that of 6, while 298 the hole mobility was improved slightly from 3.26 to 4.50 imes299 10^{-4} cm² V⁻¹ s⁻¹ after introducing a larger and rigid planar 300 conjugated BDT unit. A better PCE of 5.44%, with similar V_{oc} 301 of 0.93 V and J_{sc} of 9.77 mA/cm², but a much improved FF of 302 0.60 was obtained by using a blend of **17** and $PC_{61}BM$, a 303 similar case to compound 6. These results indicate that the 304 BDT-modified septithiophene system is a better backbone 305 system than the pristine septithiophene system. 306

Continuing with this line of thought, compound 18 using 307 2-ethylhexyloxy substituted BDT as a central unit was 308 synthesized.²⁶ Meanwhile, the more easily synthesized 309 dioctyltertthiophene instead of trioctylterthiophene was 310 used as the spacer with the electron withdrawing terminal 311 unit. For BHJ devices with PC₆₁BM as the acceptor, a mod-312 erate PCE of 4.56% for 18 with Voc of 0.95 V, FF of 0.60, and 313 $J_{\rm sc}$ of 8.00 mA/cm² was obtained. Compound **19** with a 314

longer backbone conjugation unit has also been synthe-315 sized and studied for OPV performance. Its V_{oc} is significantly 316 lower (0.79 V) than compound 18 with a shorter conjugation 317 system, corresponding to the optimized length (7 units) for 318 the oligothiophene systems above. This prompted us 319 to focus on the continuing work on systems with a 320 7-conjugated unit backbone like compound 18. From here 321 and for achieving higher PCEs, the most important issue is 322 how to improve J_{sc} while retaining high values of V_{oc} and FF. 323 Considering the high J_{sc} of compound **7** after introducing the 324 RD unit, we designed and synthesized molecules 20-23 with 325 a BDT central building block and a RD unit as the terminal 326 unit.^{8,26} As shown in Table 2, all OPV devices based on these 327 four molecules have a high J_{sc} over 11 mA/cm² and V_{oc} over 328 0.9 V. Furthermore, with larger conjugation in the orthogo-329 nal direction, 21 and 23, respectively demonstrated higher 330 PCEs of 7.51% and 7.58% with a generally improved J_{sc}. The 331 performance could further be improved by the addition of a 332 small amount of polydimethylsiloxane (PDMS) in the active 333 layer (Figure 8 for J-V of **21**),^{8,26,40} with the best PCE at 334 **F8** 8.12% (certified 7.61%). These exciting results demonstrate 335 that the same or even better solar cell performance for small 336 molecules as that for polymers could indeed be achieved 337

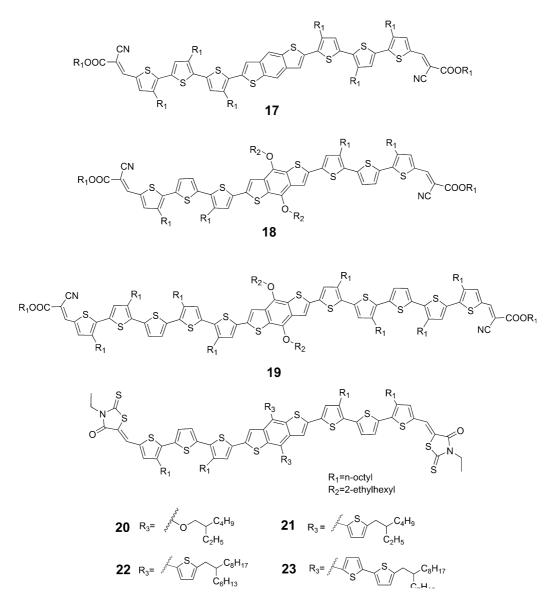


FIGURE 7. Chemical structures of molecules 17–23.

SM	$\lambda_{ m max}^{ m film}$ (nm)	$E_{\rm g}^{\rm opt}$ (eV)	$E_{\rm g}^{\rm cv}$ (eV)	HOMO (eV)	LUMO (eV)	$V_{\rm oc}$ (V)	J _{sc} (mA/cm ²)	FF	PCE (%)	re
17	563	1.83	1.57	-5.11	-3.54	0.93	9.77	0.599	5.44	25
18	560	1.84	1.80	-5.04	-3.24	0.95	8.00	0.60	4.56	26
19	581	1.79	1.73	-4.98	-3.25	0.79	9.51	0.683	5.13	С
20	583, 630	1.74	1.75	-5.02	-3.27	0.93	11.40	0.653	6.92	26
						0.93	12.21	0.65	7.38 ^a	26
21	591, 640	1.72	1.75	-5.02	-3.27	0.91	13.15	0.628	7.51	8
						0.93	13.17	0.663	8.12 ^a	8
22	589, 633	1.76	1.77	-5.06	-3.29	0.96	12.36	0.533	6.32	8
						0.96	11.92	0.594	6.79 ^a	8
23	591, 639	1.76	1.78	-5.07	-3.29	0.90	11.97	0.704	7.58	8
						0.92	12.09	0.721	8.02 ^a	8
24	596	1.73	1.69	-4.95	-3.26	0.80	11.51	0.64	5.84	27
25	720	1.50	1.60	-5.20	-3.60	0.78	14.4	0.593	6.70	7
26	678	1.55	1.78	-5.12	-3.34	0.809	12.8	0.68	7.0	41
						0.773	14.74	0.724	8.24 ^b	42

through careful molecule design and device optimization. 338 The high FF (over 0.70) and EQE (over 70%) for these 339 compounds are comparable with that of the best P-OPV 340 devices.^{4,34} It is well accepted that the preferable domain 341 size for BHJ OPV devices is in the range of tens of nm, 342 depending on each compound.^{6,15} Indeed, morphology 343 analysis by TEM and AFM indicated that a better morphol-344 ogy with a more uniform continuous interpenetrating net-345 work and preferable domain size (15–20 nm) were formed 346 after addition of PDMS to these compounds.^{8,26} 347

348 DTS Core Based Small Molecules

Parallel to the studies of BDT-modified septithiophene systems, compound **24** with a DTS unit as the central building

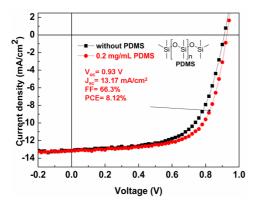


FIGURE 8. J-V curves of solar cells with an active layer composed of compound **21**:PC₇₁BM (1:0.8, w/w) without and with PDMS.

block (Figure 9) was prepared to compare with compound 351 F9 **18.**²⁷ The optimized device gave a PCE of 5.84%, with a 352 lower V_{oc} of 0.80 V, similar J_{sc} of 11.51 mA/cm², and FF of 353 0.64. The lower V_{oc} is actually consistent with the 354 literature,³⁹ showing that the DTS unit can decrease the 355 V_{oc} in many cases. 356

Bazan and co-workers reported a series of DTS-based 357 small molecules as donors for solution processed SM-OPV 358 devices.^{7,41} Typical molecules are presented in Figure 9. 359 Different from our molecules, 25 is based on an A-D-A 360 framework with donor end-capping units. Devices fabri-361 cated from the blend solutions with 25:PC71BM (7:3 w/w) 362 showed a PCE of 4.52%, with J_{sc} of 12.5 mA/cm², V_{oc} of 0.80 V, 363 and FF of 0.452. When 0.25% (v/v) DIO (1,8-diiodooctane) 364 was added during the film-forming process, a PCE of 6.7% 365 was achieved.⁷ The improvement of PCE was ascribed to the 366 formation of more preferable domain size as a result of the 367 addition of DIO. 368

In their following work, 5-fluorobenzo[d][1,2,5]thiadiazole 369 instead of [1,2,5]thiadiazolo[3,4-c]pyridine was used as the 370 acceptor unit and molecule 26 was synthesized.⁴¹ A BHJ 371 device with **26** and $PC_{71}BM$ as the active layer and annealed 372 at 130 °C gave a PCE of 5.8% with $V_{\rm oc}$ of 0.82 V, $J_{\rm sc}$ of 373 10.8 mA/cm², and FF of 0.65. When 0.4 v/v% DIO as a solvent 374 additive was incorporated, an improved PCE of 7.0% ($V_{oc} =$ 375 0.809 V, $J_{sc} = 12.8 \text{ mA/cm}^2$, and FF = 0.68) was achieved. Just 376 recently, a PCE of over 8% has been achieved with 26 and 377 PC₇₁BM as the active layer through device optimization.⁴² 378

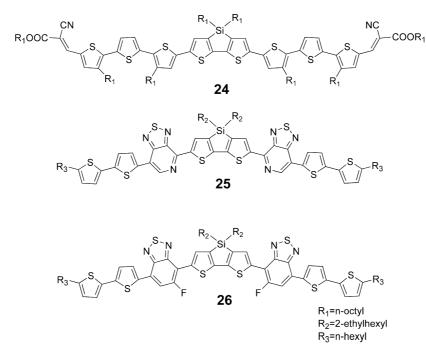


FIGURE 9. Chemical structures of molecules 24-26.

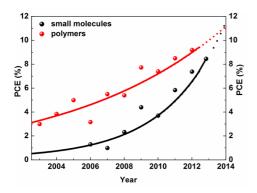


FIGURE 10. Highest PCEs of solution processed small-molecule- and polymer-based single layer OPV devices each year for the past 10 years. The details of each point, including molecule structures and PCEs, are presented in the Supporting Information (Table S1).

Conclusions and Outlook

380 It is fair to say that the studies of solution-processed SM-OPV have a short history and have been much less explored 381 compared with that for P-OPV, though the overall performance 382 is catching up and becoming close to the best obtained for 383 P-OPV, particularly in the past few years. Figure 10 summarizes 10384 the best PCEs for each of the last 10 years for both small-385 molecule- and polymer-based single layer devices. It can be 386 seen clearly that there is a great possibility that the perfor-387 388 mance of SM-OPV devices can approach or even exceed the performance of polymer devices. In fact, PCEs of over 10% 389 have been reported for a small-molecule-based device pro-390 duced industrially, though no detailed information on the 391 molecule and device structures has been reported.⁴³ 392

From eq 1, to achieve high PCEs, all the three factors need to be optimized simultaneously. For this, our approach is to take the advantage of both *conventional* polymer and small molecule materials and maximize the synergistic effect of different building blocks.

 $V_{\rm oc}$: It is now widely accepted that the $V_{\rm oc}$ is strongly 398 correlated with the energy level difference between the HOMO 399 of the donor and the LUMO of the acceptor for both small 400 molecule and polymer-based solar cells, though, generally the 401 $V_{\rm oc}$ of SM-OPV devices is higher than those of analogous 402 polymer devices. Currently, the optimized $V_{\rm oc}$ for SM-OPV 403 devices is in the range of 0.90–1.1 V.^{8,44} It appears that smaller 404 conjugation systems tend to give higher $V_{\rm oc}$ but with decreased 405 $J_{\rm sc}$. The question is then how to increase $V_{\rm oc}$ without significantly 406 decreasing the value of J_{sc} . This might be achieved by putting 407 some electron-withdrawing groups on the same molecule 408 backbone as has been observed for some polymer systems.⁴⁵ 409

410 J_{sc} : The best J_{sc} for a SM-OPV device is around 15 mA/ 411 cm².¹² This obviously needs much improvement compared with the much higher one (22 mA/cm²) for polymers.³⁶ In 412 addition to controlling and finely tuning the band gap with 413 ideal value similar to that of polymers⁶ (around 1.5 eV) by 414 conjugation length and the backbone composition to respond 415 the near-IR range sunlight, our results indicate that using dye 416 units could be an effective approach. This should have great 417 room for improvement, since the end unit, each side group, 418 and backbone units can all be finely turned individually or 419 collectively using the rich synthesis chemistry. Obviously, 420 many other approaches to improve J_{sc} could be used.^{8,10–12} 421

FF: FF is probably the most complicated factor to predict 422 and control because it depends on many and sometimes 423 competing factors. The intrinsic ones include the mobility 424 and planarity of molecules, side groups, solubility, molecular 425 packing of both donor and acceptor, donor and acceptor 426 miscibility, and so on. Some external morphology control 427 techniques can be used to optimize them, such as solvent/ 428 thermal annealing, using mixtures of solvents, extra addi-429 tives, interface layers and so on.^{4,6} Currently, FF over 0.70 430 has been achieved for some small molecules using solution 431 process.^{8,12,22,42} This indicates that as high FF for SM-OPV 432 devices could also be achieved as for P-OPV devices. One of 433 the keys issues is to achieve high mobility but avoiding the 434 serious but intrinsic intendance of molecular packing/ 435 aggregation between the donor small molecules in the 436 active layer. 437

The important lesson we have learned is that small 438 molecules for OPV should be designed by learning from 439 the development of both conventional polymers and small 440 molecules and taking the advantages from both of them 441 together. For this, conventional small molecules might not be 442 up to the task. Taking a glance at the chemical structures of 443 the highest efficiency small molecules, one can see that 444 these are not *conventional* small molecules. Some general 445 rules listed below could be useful for future design. First, a 446 balanced conjugation ($\sim 6-10$ units) is needed to to give 447 high light absorption. Second, high solubility is needed and 448 could be achieved using alkyl groups on the backbone. 449 Third, relatively good planarity is also required to achieve 450 high charge mobility but still to avoid over molecular lamel-451 lar packing of the donor molecules themselves. The fourth is 452 to control and tune the energy levels of these molecules to 453 match that of fullerene and other acceptors. The fifth, and 454 equally important, concerns the morphology and device 455 optimization beyond molecule design. Importantly, if we 456 believe that a material has an intrinsic best performance, it is 457 important to realize that the best observed performance 458 for the material is very likely not its intrinsic best one. 459

Optimization can only bring it close to the intrinsic value. With 460 this and the current best results being still far from commercial-461 ization, one must bear in mind that it is important to design 462 better molecules with intrinsic better performance. The accu-463 mulated results in the large amount of literature, including 464 many mechanism studies, may indicate that it is time to use 465 advanced modeling/theory to design and predict some opti-466 mized molecules, though this could prove a very tricky job. 467

468 Supporting Information. The details of each point in469 cluding molecule structures and PCE in Figure 10. This
470 material is available free of charge via the Internet at
471 http://pubs.acs.org.

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The authors gratefully acknowledge financial support from the
MOST (Grants 2012CB933401 and 2011DFB50300), NSFC
(Grants 51273093 and 50933003), MOE (Grant, PCSIRT
IRT1257), and NSF of Tianjin (Grants 10ZCGHHZ00600).

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487 FOOTNOTES

488 The authors declare no competing financial interest.

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